# Lone-Pair Self-Containment in Tellurite Tubes: Hydrothermal Syntheses and Structures of $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ 

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Traditionally, mixed-metal oxides containing a dense packing of atoms have been prepared in powder (microcrystalline) form by high-temperature ceramic methods. ${ }^{1}$ It has recently been shown ${ }^{2,3}$ that hydrothermal (or nonaqueous solvothermal) synthesis is a highly effective route to single crystals of new, condensed, anhydrous tellurites. Crystal structures containing $\mathrm{Te}^{\mathrm{IV}}$ are of interest ${ }^{4}$ because of the unpredictable coordination behavior shown by this species due to its stereochemically active lone pair of electrons. In fact, the entire structure is affected by the requirement for "empty" space to accommodate the tellurium lone-pair electrons. ${ }^{5,6}$ In this way in particular, they may assist in understanding the properties of technologically important tellurite glasses by rationalizing the connectivity patterns for the $\mathrm{Te} / \mathrm{O}$ units and the coordination requirements of network-modifying cations. ${ }^{7,8}$ Here, we report the mild-condition hydrothermal syntheses and singlecrystal structures of two new barium tellurites, $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BeTe}_{4} \mathrm{O}_{9}$. These related phases contain novel, infinite, tellurite tubes and may be regarded as two members of the family of phases denoted $\mathrm{BaTe}_{n} \mathrm{O}_{2 n+1}$ with $n=3$ and 4. A ceramic synthesis for $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ was reported some time ago, ${ }^{9}$ but no crystallographic details were elucidated.
$\mathrm{BaTe}_{3} \mathrm{O}_{7}$ was prepared from $\mathrm{Ba}(\mathrm{OH})_{2} \cdot 8 \mathrm{H}_{2} \mathrm{O}(0.318 \mathrm{~g}, 1 \mathrm{mmol})$, $\mathrm{TeO}_{2}(0.638 \mathrm{~g}, 4 \mathrm{mmol})$, and 10 mL of $\mathrm{H}_{2} \mathrm{O}$, which were placed in a $18-\mathrm{mL}$ capacity Teflon-lined steel bomb. Dropwise addition of concentrated $\mathrm{HNO}_{3}$ lowered the starting pH from 12 to $\sim 10$ and the resulting mixture was heated to $180^{\circ} \mathrm{C}$ for 5 days. After removal from the oven and cooling over a few hours, the bomb was opened and the solid product, consisting of 0.62 g of transparent rods (yield based on $\mathrm{Ba}=98 \%$ ), was recovered by vacuum filtration and washing with water and acetone. Reactions carried out at pH 12 (no $\mathrm{HNO}_{3}$ in the reaction) led to as-yet unidentified powder products.

Single crystals of $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ were prepared from a mixture of $\mathrm{BaCO}_{3}(0.199 \mathrm{~g}, 1 \mathrm{mmol}), \mathrm{TeO}_{2}(0.477 \mathrm{~g}, 3 \mathrm{mmol}), \mathrm{CoCl}_{2} \cdot 6 \mathrm{H}_{2} \mathrm{O}$ ( $0.238 \mathrm{~g}, 1 \mathrm{mmol}$ ), and 10 mL of $\mathrm{H}_{2} \mathrm{O}$. The starting pH was adjusted to $\sim 10$ by the addition of 1 M LiOH solution and the sealed mixture was heated to $180^{\circ} \mathrm{C}$ for 3 days. Product recovery as above led to a mixture of pinkish-purple powder, probably $\mathrm{Co}(\mathrm{OH})_{2}$, and a few transparent rods and bars of $\mathrm{BaTe}_{4} \mathrm{O}_{9}$. We have not yet managed to prepare $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ in high yield by hydrothermal methods. Similar syntheses carried out in the absence of cobalt chloride, which possibly controls the pH of the reaction, led only to recrystallized $\mathrm{TeO}_{2}$. Pure polycrystalline $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ can be made by ceramic methods: a stoichiometric mixture of $\mathrm{BaCO}_{3}$ and $\mathrm{TeO}_{2}$ heated to $550{ }^{\circ} \mathrm{C}$ for 3 days with intermediate re-grindings led to pure $\mathrm{BaTe}_{4} \mathrm{O}_{9}$.

The X-ray powder patterns (Philips PW1710 diffractometer, Cu $\mathrm{K} \alpha$ radiation, $\lambda=1.5418 \AA, T=25 \pm 2{ }^{\circ} \mathrm{C}$ ) of hydrothermally

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Figure 1. Polyhedral view down [010] of $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ with the tellurium centers forming one $\left[\mathrm{Te}_{3} \mathrm{O}_{7}\right]^{2-}$ tube labeled. The empty tube is indicated by E .
made $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and ceramic $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ were in excellent agreement with simulations based on the single-crystal structures, indicating phase purity and a high degree of crystallinity. TGA for $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ indicated no weight loss until $\sim 850{ }^{\circ} \mathrm{C}$, followed by continuous weight loss to $1000^{\circ} \mathrm{C}$, the limit of operation of the instrument. The IR spectra of $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ were consistent with their anhydrous nature and previous data for tellurites. ${ }^{10}$

Suitable single crystals of $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ (transparent rod, $0.26 \times$ $0.04 \times 0.04 \mathrm{~mm}$ ) and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ (transparent bar, $0.36 \times 0.06 \times$ 0.03 mm ) were mounted on thin glass fibers and their structures were determined by standard X-ray methods ${ }^{11}$ using a Bruker SMART1000 CCD diffractometer.
$\mathrm{BaTe}_{3} \mathrm{O}_{7}$ contains 11 unique atoms ( $1 \mathrm{Ba}, 3 \mathrm{Te}, 7 \mathrm{O}$ ), all of which occupy general positions. The oxygen atom geometries around Te 1 and Te 2 are well described as $\mathrm{TeO}_{4}$ folded squares, as seen previously for similar groupings. ${ }^{2}$ This geometry can be simply rationalized in VSEPR theory as an $\mathrm{AX}_{4} \mathrm{E}$ trigonal bipyramid, with the lone pair occupying an equatorial position. ${ }^{12} \mathrm{Te} 3$ has less regular coordination, with three short $(<2.00 \AA)$, one intermediate length $(2.24 \AA)$, and one very long $(2.56 \AA) \mathrm{Te}-\mathrm{O}$ bond. The overall shape of this $\mathrm{TeO}_{3+1+1}$ grouping approximates to a distorted squarebased pyramid. Bond valence sum (BVS) ${ }^{13}$ values of 4.02, 4.05, and 4.15 for $\mathrm{Te} 1, \mathrm{Te} 2$, and Te 3 , respectively, are in good agreement with the expected 4.00. Six O atoms participate in $\mathrm{Te}-\mathrm{O}-\mathrm{Te}$ bridges $\left(\theta_{\mathrm{av}}=116.1^{\circ}\right)$ and one is terminal to Te1. All the O atoms also bond to one or more of the 11-coordinate barium cations $\left[d_{\mathrm{av}}(\mathrm{Ba}-\mathrm{O})=2.997(3) \AA\right]$

The polyhedral connectivity in $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ involves both vertexand edge-sharing of the tellurite moieties. The Te1-, Te2-, and Te3centered groups form 6-ring loops by way of $\mathrm{Te}-\mathrm{O}-\mathrm{Te}$ bridges (sequence: Te1, Te2, Te3, Te1, Te2, Te3). Stacking of these loops in the $[010]$ direction leads to distinctive macroanionic $\left[\mathrm{Te}_{3} \mathrm{O}_{7}\right]^{2-}$ tubes (Figure 1). Despite their atom-to-atom dimensions of $\sim 4.39 \times 5.90 \AA$, they are completely empty. Although it is difficult


Figure 2. Polyhedral view down [010] of $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ with the tellurium centers forming one $\left[\mathrm{Te}_{4} \mathrm{O}_{9}\right]^{2-}$ tube labeled. The empty tube is indicated by E.
to locate $\mathrm{Te}^{\mathrm{IV}}$ lone pairs of electrons for non-pyramidal geometries, ${ }^{14}$ it appears here that the Te 1 and Te 2 lone pairs point into the inside of the tube. Finally, the $\left[\mathrm{Te}_{3} \mathrm{O}_{7}\right]^{2-}$ tubes are cross-linked by way of edge-sharing through pairs of $\mathrm{Te} 3-\mathrm{O} 7-\mathrm{Te} 3^{\prime}$ bonds to result in complex anionic sheets propagating in the (101) planes. The chargebalancing barium cations occupy inter-sheet sites.
$\mathrm{BaTe}_{4} \mathrm{O}_{9}$ contains 14 unique atoms ( $1 \mathrm{Ba}, 4 \mathrm{Te}, 9 \mathrm{O}$ ), all on general positions. Te 1 and Te 4 have $(3+1) \mathrm{O}$ atom coordination, similar to the situation ${ }^{2}$ in $\mathrm{Co}_{2} \mathrm{Te}_{3} \mathrm{O}_{8} . \mathrm{Te} 2$ and Te 3 have four O atom neighbors in folded square conformations, with the axial $\mathrm{Te}-\mathrm{O}$ bonds showing their usual lengthening relative to the equatorial bonds. BVS values of $3.90,3.97,3.96$, and 3.87 result for $\mathrm{Te} 1, \mathrm{Te} 2, \mathrm{Te} 3$, and Te 4 , respectively. Five O atoms participate in $\mathrm{Te}-\mathrm{O}-\mathrm{Te}$ bridges $\left(\theta_{\mathrm{av}}=120.1^{\circ}\right)$, three are terminal to Te , and one (O7) is three-coordinate to Te neighbors. All the O atoms except $\mathrm{O} 2, \mathrm{O}$, and O 8 also bond to one or more barium cations, which are nine coordinate $\left[d_{\mathrm{av}}=2.904\right.$ (3) $\AA$ ] in this structure.

The packing of the polyhedral building units in $\mathrm{BeTe}_{4} \mathrm{O}_{9}$ can be visualized in terms of infinite tubes of stoichiometry $\left[\mathrm{Te}_{4} \mathrm{O}_{9}\right]^{2-}$ propagating along the $b$ direction. In this case, the tubes are built up from 8 -ring loops (sequence: $\mathrm{Te} 1, \mathrm{Te} 2, \mathrm{Te} 3, \mathrm{Te} 4, \mathrm{Te} 1, \mathrm{Te} 2$, Te3, Te4). Stacking of the rings along the short $b$ unit cell direction leads to infinite tellurite tubes (Figure 2). These units are highly squashed, with atom-to-atom dimensions of $3.49 \times 7.97 \AA$, and the Te 2 and Te 3 lone pairs appear to point into the tubes. As in $\mathrm{BaTe}_{3} \mathrm{O}_{7}$, the $\left[\mathrm{Te}_{4} \mathrm{O}_{9}\right]^{2-}$ tubes in $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ are cross-linked by edgesharing. In $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ the edge-sharing occurs by way of $\mathrm{Te} 4-\mathrm{O} 7-$ $\mathrm{Te} 4^{\prime}$ double bridges in the $c$ direction and $\mathrm{Te} 1-\mathrm{O} 1-\mathrm{Te} 1^{\prime}$ double bridges in the $a$ direction to result in a three-dimensional $\mathrm{Te} / \mathrm{O}$ network encapsulating the barium cations in [010] tunnels.

In summary, the title compounds demonstrate not just typical local distortions ${ }^{5}$ of $\mathrm{Te} / \mathrm{O}$ groupings, but a novel way to accom-
modate $\mathrm{Te}^{\mathrm{IV}}$ lone-pair electrons in infinite, "self-contained" onedimensional tubes. In fact, these tubes appear to be large enough to accommodate other chemical species such as $\mathrm{H}^{+}$or $\mathrm{Li}^{+}$, which suggests the possibility of redox intercalation and/or ion-exchange chemistry for this type of structure. These processes seem unlikely in the title compounds where there are no easily reducible cations in the framework. However, extended tellurite networks incorporating species such as $\mathrm{Mn}^{3+}$ or $\mathrm{Fe}^{3+}$ might allow for the incorporation of lithium with concomitant reduction of $\mathrm{M}^{3+}$ to $\mathrm{M}^{2+}$ and we are continuing to investigate such systems.

Some time ago, Wroblewska et al. ${ }^{9}$ prepared several barium tellurites by ceramic methods, but few structural details were elucidated. The unindexed powder pattern reported for $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ (JCPDS card 36-0887) corresponds reasonably well to our data for the same phase. Interestingly, the phase $\mathrm{BaTe}_{2} \mathrm{O}_{5}$ reported by the same workers, which represents the $n=2$ member of the $\mathrm{BaTe}_{n} \mathrm{O}_{2 n+1}$ series, is possibly ${ }^{15}$ monoclinic, with unit cell dimensions of $a=19.85 \AA, b=4.38 \AA, c=7.39 \AA$, and $\beta=95.7^{\circ}$ (JCPDS card 36-0886). This cell shape is similar to the "squashed" (small b) monoclinic cells found here for $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ and perhaps indicates that the putative $\mathrm{BaTe}_{2} \mathrm{O}_{5}$ possesses a similar structure to the title compounds. We are investigating the $n=2$ material and endeavoring to prepare further members of this unusual family such as the $n=5$ phase $\mathrm{BaTe}_{5} \mathrm{O}_{11}$.

Supporting Information Available: Crystallographic data for $\mathrm{BaTe}_{3} \mathrm{O}_{7}$ and $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ (CIF). This material is available free of charge via the Internet at http://pubs.acs.org.

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(11) Crystal data for $\mathrm{BaTe}_{3} \mathrm{O}_{7}: M_{\mathrm{r}}=632.14$, monoclinic, $P 2_{1} / n$ (No. 14), $a=10.2765$ (6) $\AA, b=5.2510$ (3) $\AA, c=13.5499$ (8) $\AA, \beta=98.980$ $(1)^{\circ}, Z=4, R(F)=0.030, w R 2\left(F^{2}\right)=0.071$. Crystal data for $\mathrm{BaTe}_{4} \mathrm{O}_{9}$ : $M_{\mathrm{r}}=791.72$, monoclinic, $C 2 / c$ (No. 15), $a=19.5354$ (10) $\AA, b=4.5099$ (2) $\AA, c=23.3021$ (12) $\AA, \beta=112.961(1)^{\circ}, Z=4, R(F)=0.039$, $w R 2\left(F^{2}\right)=0.095$.
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